

Journal of  
**Micro/Nanolithography,  
MEMS, and MOEMS**

Nanolithography.SPIEDigitalLibrary.org

# **Optimization of directed self-assembly hole shrink process with simplified model**

Kenji Yoshimoto  
Ken Fukawatase  
Masahiro Ohshima  
Yoshihiro Naka  
Shimon Maeda  
Satoshi Tanaka  
Seiji Morita  
Hisako Aoyama  
Shoji Mimotogi

# Optimization of directed self-assembly hole shrink process with simplified model

Kenji Yoshimoto,<sup>a,b,\*</sup> Ken Fukawatase,<sup>b</sup> Masahiro Ohshima,<sup>b</sup> Yoshihiro Naka,<sup>c</sup> Shimon Maeda,<sup>c</sup> Satoshi Tanaka,<sup>c</sup> Seiji Morita,<sup>c</sup> Hisako Aoyama,<sup>c</sup> and Shoji Mimotogi<sup>c</sup>

<sup>a</sup>Kyoto University, Center for the Promotion of Interdisciplinary Education and Research, Sakyo-ku, Kyoto 606-8501, Japan

<sup>b</sup>Kyoto University, Department of Chemical Engineering, Katsura Campus, Nishikyō-ku, Kyoto 615-8510, Japan

<sup>c</sup>Toshiba Corporation, 1 Komukai Toshiba, Saiwai, Kawasaki 212-8583, Japan

**Abstract.** Application of the directed self-assembly of block copolymer to the hole shrink process has gained large attention because of the low cost and high potential for sublithographic patterning. In this study, we have employed a simplified model, called the Ohta-Kawasaki model to find the optimal process conditions, which minimize the morphological defects of the diblock copolymer, PS-*b*-PMMA. The model parameters were calibrated with cross-sectional transmission electron microscopy images. Our simulation results revealed that it is difficult to eliminate the morphological defects (i.e., PS residual layer) by only varying the shape of the guide hole. It turned out that changing the affinity of the bottom surface of the guide hole from “PMMA attractive” to “neutral” is a more effective way to obtain a reasonably wide, defect-free process window. Note that our simulations are not only computationally inexpensive, but are also comparable to the other detailed models such as the self-consistent field theory; they may also be feasible for large-scale simulations such as the hotspot analysis over a large area. © The Authors. Published by SPIE under a Creative Commons Attribution 3.0 Unported License. Distribution or reproduction of this work in whole or in part requires full attribution of the original publication, including its DOI. [DOI: [10.1117/1.JMM.13.3.031305](https://doi.org/10.1117/1.JMM.13.3.031305)]

Keywords: directed self-assembly; block copolymer; simulation; model; defects; hole shrink.

Paper 14055SSP received Apr. 16, 2014; revised manuscript received Jul. 10, 2014; accepted for publication Jul. 15, 2014; published online Aug. 7, 2014.

## 1 Introduction

The directed self-assembly (DSA) has been considered as a viable option for next generation lithography. Some DSA patterning processes have already been investigated in a manufacturing environment, e.g., density multiplication of line (or hole) patterns,<sup>1</sup> and hole shrink process.<sup>2</sup> One of the remaining issues for these DSA processes is to reduce the defect level.<sup>3,4</sup> According to the International Technology Roadmap for Semiconductors (ITRS), the defect level for the current DSA processes is still far from the manufacturing requirement.<sup>5</sup>

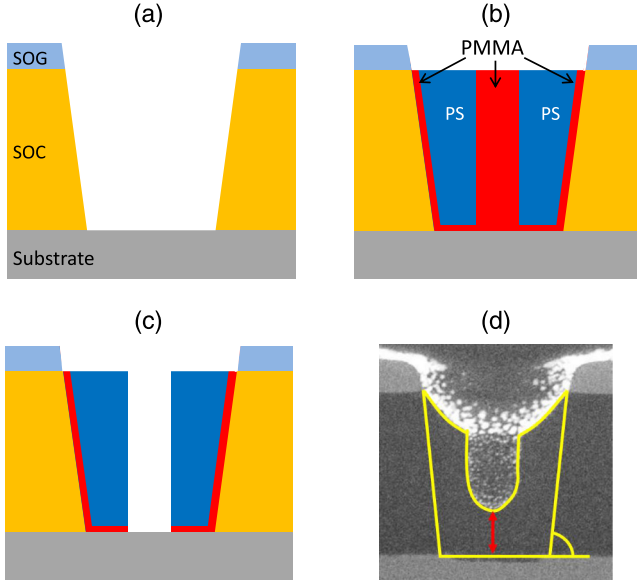
The DSA hole shrink process which we have investigated in this study is implemented as follows (Fig. 1).<sup>2</sup> First, the guide holes (e.g., contact or via holes) are etched into the silicon-on-glass (SOG) and silicon-on-carbon (SOC) hard-masks, using conventional lithographic methods [Fig. 1(a)]. Then, an asymmetric poly(styrene-block-methylmethacrylate) diblock copolymer (PS-*b*-PMMA) is spin-coated and annealed on the etched wafer. Here, the minor block is PMMA and the major block is PS. After annealing, the PS-*b*-PMMA is phase-separated into the PS-rich and PMMA-rich domains, and the PMMA-rich domain is transformed into a vertical cylinder [Fig. 1(b)]. Note that the sidewalls and bottom surface of the guide hole are covered with a very thin layer of PMMA due to their higher affinity to the PMMA.<sup>2,6</sup> The diameter of the cylindrical PMMA domain is typically on the order of ~20 to 30 nm, depending on the molecular weight of the PS-*b*-PMMA. After the

selective removal of the cylindrical PMMA domain, the smaller hole is generated in the guide hole pattern.

A crucial issue for the DSA hole shrink process is the morphological defects of the block copolymer generated during the annealing step.<sup>2,4</sup> As shown in Fig. 1(d), the relatively thick PS layer remains between the cylindrical PMMA domain and the bottom surface. Depending on the thickness of the PS residual layer, it will be difficult to control the etching process which is followed by the DSA hole shrink process. It is also important to note that the PS residual layer is hardly detected by the conventional top-down scanning electron microscope (SEM) image.

Simulation can be a powerful tool to predict the three-dimensional (3-D) morphology of PS-*b*-PMMA in the guide hole. Some simulation methods, such as dissipative particle dynamics (DPD)<sup>6,7</sup> and self-consistent field theory (SCFT)<sup>8</sup> have successfully demonstrated the formation of the cylindrical PMMA domain and the bottom PS layer with the sidewalls and bottom surface attractive to the PMMA. These simulation methods could also provide some useful insights into the chain conformation of the block copolymer, although they would be computationally expensive, particularly for the 3-D systems required for the DSA hole shrink process. In this study, we have employed a simplified model, called the Ohta-Kawasaki (OK) model<sup>9,10</sup> to achieve a quick turnaround time for finding the optimal process condition for the DSA hole shrink process. In prior to the optimizations, we have calibrated the model parameters (i.e., interactive parameters between the walls of the guide hole and the blocks of PS-*b*-PMMA) with cross-section transmission electron microscopy (TEM) images. Taking advantage of the considerably small computational expense of the OK

\*Address all correspondence to: Kenji Yoshimoto, E-mail: [yoshimoto1@cheme.kyoto-u.ac.jp](mailto:yoshimoto1@cheme.kyoto-u.ac.jp)



**Fig. 1** Directed self-assembly hole shrink process: (a–c) schematic process flow and (d) cross-sectional transmission electron microscopy (TEM) image after the removal of the cylindrical PMMA domain. In (d), the yellow line illustrates the outline of PS domain, the red arrow indicates the thickness of the bottom PS layer.

model, in the optimizations we have explored various guide hole shapes and minimized the thickness of the PS residual layer.

This paper is organized as follows. Section 2 describes the details of the simplified model used in this study. Section 3 discusses our simulation results, including the comparison of the OK model with the SCFT (Sec. 3.1), calibration of the model parameters with the cross-sectional TEM images (Sec. 3.2), minimization of the thickness of the PS residual layers (Sec. 3.3), and the model extensibility to multiple cylinder cases (Sec. 3.4).

## 2 Model and Methods

### 2.1 Model

In this study, we employed the OK model described in Ref. 9. The free energy of the bulk diblock copolymer,  $F_b$ , is expressed as

$$F_b\{\eta(\mathbf{r})\} = \left\{ -\frac{\tau}{2}\eta(\mathbf{r})^2 + \frac{g}{4}\eta(\mathbf{r})^4 + \frac{1}{2}[\nabla\eta(\mathbf{r})]^2 + \frac{\alpha}{2} \int d\mathbf{r}' G(\mathbf{r}, \mathbf{r}') [\eta(\mathbf{r}) - \bar{\eta}] [\eta(\mathbf{r}') - \bar{\eta}] \right\}, \quad (1)$$

where the coefficients,  $\tau$ ,  $g$ , and  $\alpha$ , are the model parameters, and  $G(\mathbf{r}, \mathbf{r}')$  is the Green function ( $\mathbf{r}, \mathbf{r}'$ : position vector). The order parameter,  $\eta(\mathbf{r})$ , is defined by  $2\phi(r) - 1$ , with  $\phi(r)$  as the volume fraction of the minor segments (i.e., PMMA). The average order parameter,  $\bar{\eta}$ , is calculated from  $2f - 1$ , where  $f$  is the fraction of the minor block per copolymer chain. Based on the OK's phase diagram,<sup>9</sup> the model parameters were selected as  $\tau = 2.8$  and  $\alpha = g = 1$ , for which a stable cylindrical morphology was formed in the bulk at  $f = 0.3$ .

The sidewalls and bottom surface of the guide are covered with a thin PMMA layer due to their higher affinity to the PMMA.<sup>2,6</sup> In order to describe such attractive interactions between the walls and the PMMA, we added the following energy term

$$F_s\{\eta(\mathbf{r})\} = \int d\mathbf{r} \Lambda(\mathbf{r}) [\phi(\mathbf{r}) - \phi_s]^2, \quad (2)$$

where  $\phi_s$  is the volume fraction of the PMMA in the vicinity of the walls. Equation (2) constrains the volume fraction of PMMA on the walls to be a constant,  $\phi_s$ . The position-dependent coefficient,  $\Lambda(\mathbf{r})$ , was set at  $\Lambda_s (>0)$  on the walls, and 0 elsewhere. (Note that the Dirichlet boundary condition, i.e.,  $\phi(\mathbf{r}) = \phi_s$  at the walls, was not sufficient to reproduce the flat and thin layer of the PMMA on the walls.<sup>11</sup>)

### 2.2 Simulation Methods

The time-dependent Ginzberg–Landau equation was employed to obtain an equilibrated morphology of the PS-*b*-PMMA in the guide hole;<sup>10</sup>

$$\frac{\partial \eta(\mathbf{r}, t)}{\partial t} = L \nabla^2 \mu, \quad (3)$$

where  $t$ ,  $L$ , and  $\mu$  are the time, diffusion constant, and chemical potential, respectively. The chemical potential,  $\mu$ , was calculated from

$$\mu = -\tau\eta(\mathbf{r}, t) + g\eta(\mathbf{r}, t)^3 - \nabla^2\eta(\mathbf{r}, t) + \alpha \int d\mathbf{r}' G(\mathbf{r}, \mathbf{r}') [\eta(\mathbf{r}') - \bar{\eta}] + \Lambda(\mathbf{r}) [\phi(\mathbf{r}) - \phi_s]. \quad (4)$$

Equation (4) was derived from the functional derivative of the system's free energy with respect to the order parameter, i.e.,  $\delta[F_b + F_s]/\delta\eta(\mathbf{r}, t)$ . After substituting Eq. (4) into Eq. (3), the time evolution equation of the order parameter was given by

$$\eta(\mathbf{r}, t + \Delta t) \approx \eta(\mathbf{r}, t) + L\Delta t \{ \nabla^2 \mu_s(\mathbf{r}, t) - \alpha [\eta(\mathbf{r}, t) - \bar{\eta}] \}, \quad (5)$$

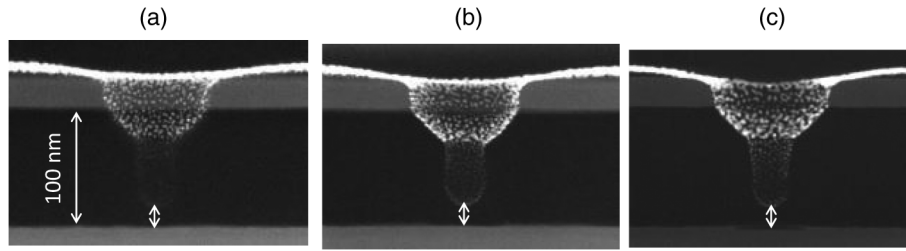
with

$$\mu_s(\mathbf{r}, t) = -\tau\eta(\mathbf{r}, t) + g\eta(\mathbf{r}, t)^3 - \nabla^2\eta(\mathbf{r}, t) + \Lambda(\mathbf{r}) [\phi(\mathbf{r}, t) - \phi_s(\mathbf{r})]. \quad (6)$$

Equations (5) and (6) were numerically resolved with the same approach which was employed in our previous study:<sup>10</sup>

**Step 1: Initialization.** Simulation box was subdivided into cubic grids. The grid size,  $\Delta l$ , was varied depending on the system size (see Table 2). The time step,  $\Delta t$ , was determined by trial and error to stabilize the iterative calculations and to maximize the calculation speed. Empirically, our calculations were stabilized at  $\Delta t/\Delta l^4 < \sim 0.0125$ .

**Step 2: Calculation of  $\mu_s(\mathbf{r}, t)$  from Eq. (6).** First, a random uniform number between  $\bar{\eta} - 0.05$  and  $\bar{\eta} + 0.05$  was assigned to each grid of the block copolymer as an initial value of the order parameter. Then, the second derivative



**Fig. 2** Cross-sectional TEM images after the removal of the cylindrical PMMA domain: (a) Sample 1, (b) Sample 2, and (c) Sample 3. See Table 1 for their details.

term,  $\nabla^2\eta(\mathbf{r}, t)$ , on the right-hand side of Eq. (6) was calculated by the standard central difference scheme.

Step 3: *Update of  $\eta(\mathbf{r}, t)$  through Eq. (5)*. Similar to Step 2, the second derivative term,  $\nabla^2\mu_s(\mathbf{r}, t)$  in the right-hand side of Eq. (5), was estimated by the standard central difference methods.

Step 4: *Iterative calculations*. Steps 2 and 3 were repeated until the maximum difference in  $\phi(\mathbf{r})$  between the current and the previous iterations became less than  $0.5 \times 10^{-6}$ . In most of our simulations, the morphology remained unchanged once all the local volume fractions were satisfied with  $|\phi(\mathbf{r}, t + \Delta t) - \phi(\mathbf{r}, t)| \leq 5 \times 10^{-6}$ .

### 2.3 Experiments and Calibration Data

The DSA hole shrink process was implemented with the same process as described in Ref. 2. (1) The SOG and the SOC were deposited on the silicon with the thickness of 35 and 100 nm, respectively. (2) The holes were etched into the SOG and the SOC hardmasks. (3) The

asymmetric PS-*b*-PMMA diblock copolymers (natural period  $L_0 = \sim 45$  nm,  $f = 0.3$ ) were spin-coated and annealed on the prepatterned substrate. (4) The cylindrical PMMA domain was formed at the center of the prepatterned hole and it was selectively removed by etching. (5) The inner hole was transferred to the substrate via plasma etching.

The cross-sectional TEM images were taken after Step (2) to estimate the 3-D size of guiding holes. The top and bottom critical dimensions (CDs) were directly measured from the TEM images, and the taper angle was estimated from the two measured CDs (assuming that the SOC thickness is  $\sim 100$  nm). Next, the cross-sectional TEM images were also taken after Step (4), i.e., selective removal of the cylindrical PMMA domain, to measure the thicknesses of the PS residual layer and the SOC layer (Fig. 2). Due to shrinking and/or tilting, there was a mismatch between the measured and actual SOC thickness ( $\sim 100$  nm); the PS residual thickness was scaled with a factor of  $(100 \text{ nm})/[\text{the measured SOC thickness (nm)}]$ . The measurement data is summarized in Table 1.

In addition to the thickness of the PS residual layer, the size of the cylindrical domain could also be measured from the cross-sectional TEM image and used as calibration metrics. However, due to some experimental uncertainties (e.g., shrinkage of the samples, change in size of the cylindrical domain before/after the removal of PMMA), it was not used for the calibration in this study.

**Table 1** Guide hole shape and thickness of the PS residual layer measured/estimated from the cross-sectional TEM images in Fig. 2.

Measured/estimated data	Sample 1	Sample 2	Sample 3
Top CD (Guide hole without BCP) (nm)	83.4	84.0	94.3
Bottom CD (Guide hole without BCP) (nm)	69.4	70.4	71.8
Taper angle (Guide hole without BCP) (deg)	86.0	86.1	83.6
PS Residual thickness (nm)	18.2	17.5	16.2

## 3 Results and Discussions

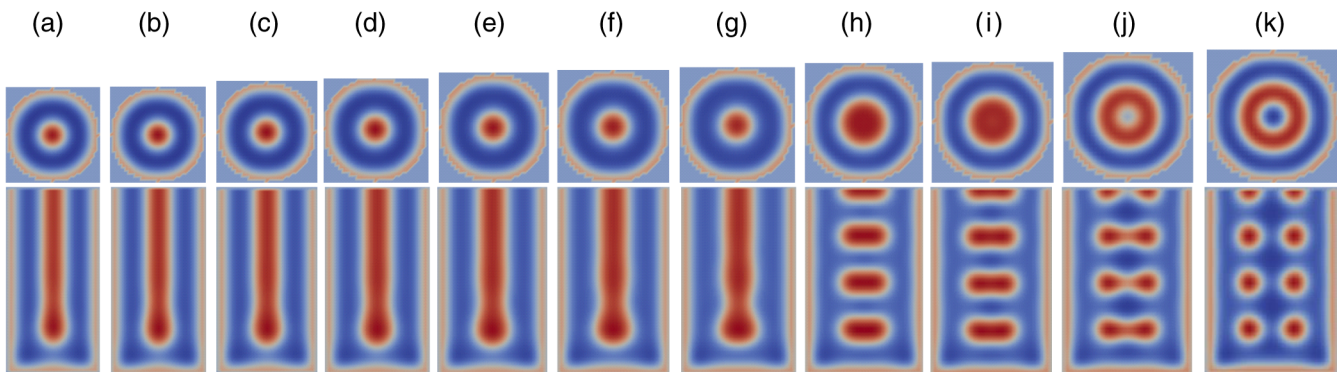
### 3.1 Three-Dimensional Morphology of PS-*b*-PMMA in Cylindrical Guide Hole

At first, morphology of the PS-*b*-PMMA in the cylindrical hole was simulated for several different hole diameters,  $D$ . The detailed simulation conditions are summarized in Table 2. The lengths are scaled by the bulk native pitch of the cylindrical PMMA domain,  $L_0$ . The surface parameters,  $\Lambda_s$  and

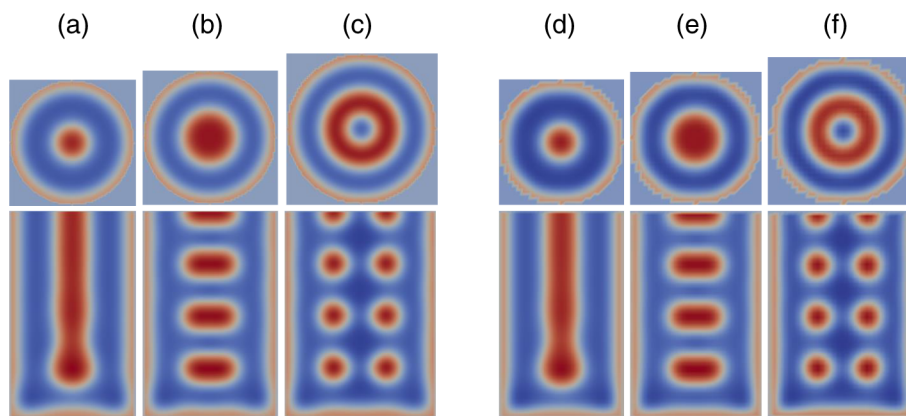
**Table 2** Simulation setup and calculation time used for the results in Fig. 3.

Diameter/ $L_0$	1.59	1.66	1.73	1.80	1.86	1.93	2.00	2.07	2.14	2.20	2.27
Grid # ( $x, y, z$ )	$34 \times 34 \times 63$	$34 \times 34 \times 61$	$34 \times 34 \times 58$	$34 \times 34 \times 56$	$34 \times 34 \times 54$	$34 \times 34 \times 52$	$34 \times 34 \times 51$	$34 \times 34 \times 49$	$34 \times 34 \times 48$	$34 \times 34 \times 46$	$34 \times 34 \times 44$
Time step ( $\tau \times 10^3$ )	0.4	0.5	0.5	0.6	0.7	0.8	1.0	1.0	1.2	1.4	1.6
Calculation time (s)	23	18	17	12	14	18	40	37	46	46	27





**Fig. 3** Morphology of PS-b-PMMA in the guide hole simulated with the Ohta–Kawasaki model: (top) top-down view and (bottom) cross-sectional view. The PS-rich and PMMA-rich domains are shown in blue and red colors, respectively. The diameter of the guide hole is increased from (a)  $1.59 [L_0]$  to (k)  $2.27 [L_0]$ , with the increment of  $\sim 0.07 [L_0]$ .



**Fig. 4** Effect of the grid size on the simulated morphologies: (a–c) 30 grids and (d–f) 76 grids in diameter. For both grid sizes, three hole diameters were selected: (a, d)  $2.00 [L_0]$ , (b, e)  $2.14 [L_0]$ , and (c, f)  $2.27 [L_0]$ .

$\phi_s$ , were set at 100.0 and 0.7, respectively, with the sidewalls and bottom surface of the cylindrical hole PMMA being strongly attracted to the side and bottom walls.

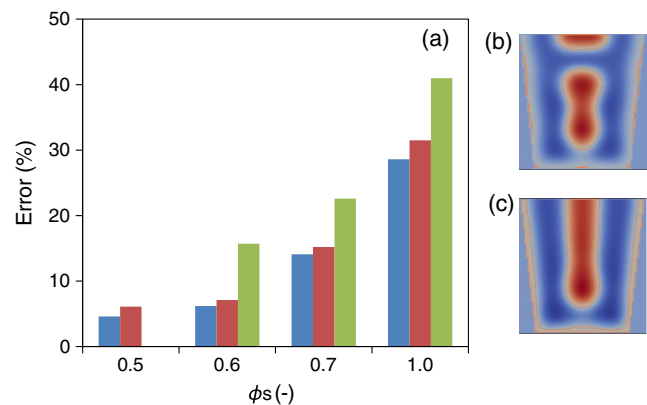
The simulation results are summarized in Fig. 3. At  $D = 1.59 - 2.00 [L_0]$ , the cylindrical PMMA domain (red color) was formed at the center of the guide hole, whereas it was not in contact with the bottom surface due to the PS layer (blue). At  $D > \sim 2.00 [L_0]$ , the cylindrical PMMA domain was transformed to the horizontal disk-shape, then to a ring-shaped morphology. Similar morphological transformations were observed by the SCFT.<sup>8</sup> It is still remarkable to see the formation of these nonbulk-like morphologies with the simplified model. It is also worth noting that each simulation was completed within 1 min (Table 2, bottom row). Such a short calculation time can be a large advantage over the other detailed models such as DPD and SCFT.

There were not any significant differences in the morphology observed by increasing the grid size (Fig. 4). It should also be noted that our numerical calculations were performed by the standard central difference method; the calculation time could be further decreased by selecting the other numerical methods (e.g., time-implicit method<sup>12</sup>).

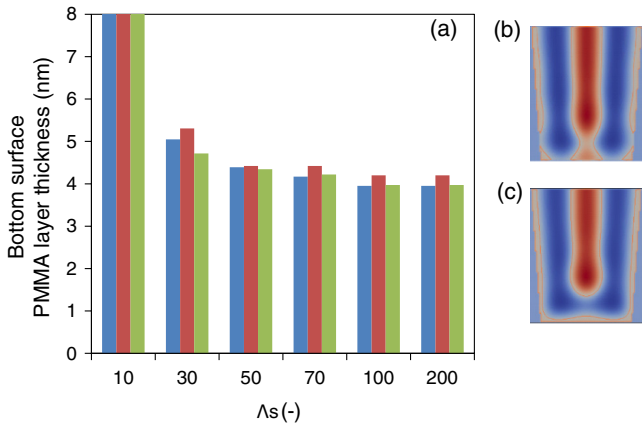
### 3.2 Model Calibration

The model calibration was implemented using the following steps. (1) The shape of the guide hole was set to be the same

dimension as the cross-sectional TEM images (Table 1). (2) Simulations were performed with the OK model. (3) The thickness of the bottom PS layer was estimated for all three samples. (4) In order to reduce the errors between the simulated thickness and the experimental data (Table 1, bottom row), the two parameters in Eq. (2),  $\phi_s$  and  $\Lambda_s$ , were adjusted.

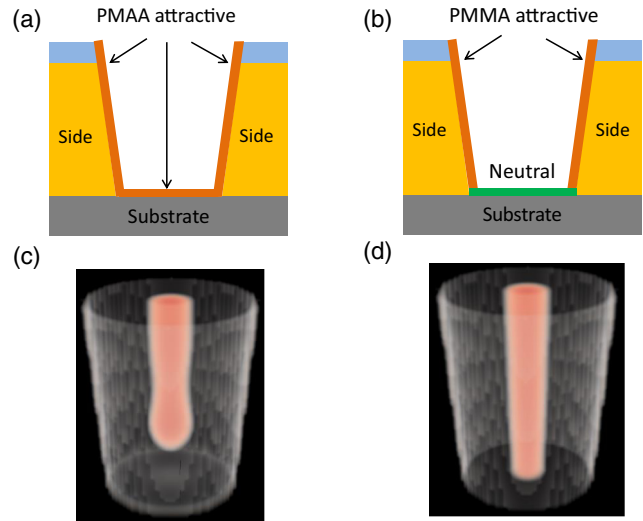


**Fig. 5** Model calibration with  $\phi_s$  ( $\Lambda_s = 100$ ). (a) Errors versus  $\phi_s$ . The errors were estimated from the difference in the thickness of the bottom PS layer between the simulations and the experiments. The colors correspond to different samples: sample 1 (blue), sample 2 (red), and sample 3 (green). (b and c) The cross-sectional morphologies of sample 3 at  $\phi_s = 0.5$  and  $\phi_s = 0.6$ , respectively.



**Fig. 6** Model calibration with  $\Lambda_s$  ( $\phi_s = 0.6$ ). (a) Relationship between the parameter  $\Lambda_s$  and the thickness of PMMA layer on the bottom surface. The colors correspond to different samples: sample 1 (blue), sample 2 (red), and sample 3 (green). (b and c) The cross-sectional morphologies of sample 1 at  $\Lambda_s = 10$  and  $\Lambda_s = 100$ , respectively.

First,  $\Lambda_s$  was fixed at 100 (sufficiently large) and  $\phi_s$  was changed from 0.5 to 1.0 (Fig. 5). As  $\phi_s$  was decreased from 1.0, the errors were monotonically decreased. At  $\phi_s = 0.5$ , however, the simulated morphology of Sample 3 was no longer cylindrical (left inset of Fig. 5); the error became the smallest at  $\phi_s = \sim 0.6$ . Next,  $\phi_s$  was fixed at 0.6 and  $\Lambda_s$  was varied from 10.0 to 200.0. It was found that the parameter,  $\Lambda_s$ , had little effect on the shape of the cylindrical domain, but had a large effect on the shape and thickness of the PMMA layer on the bottom surface (Fig. 6). At  $\Lambda_s = 10$ , the PMMA layer on the bottom surface was undulating and touching the cylindrical PMMA domain (left inset of Fig. 6).

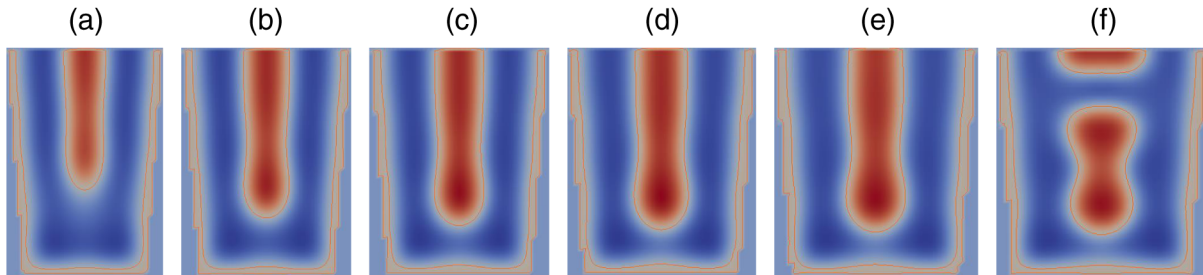


**Fig. 9** Effect of the affinity of the bottom surface on the morphology of the cylindrical PMMA domain: (a and c) PMMA-attractive, and (b and d) neutral bottom surface. The schematic cross-section of the guide hole is illustrated on the top, and the simulated morphology of the cylindrical PMMA domain (orange color) is shown on the bottom.

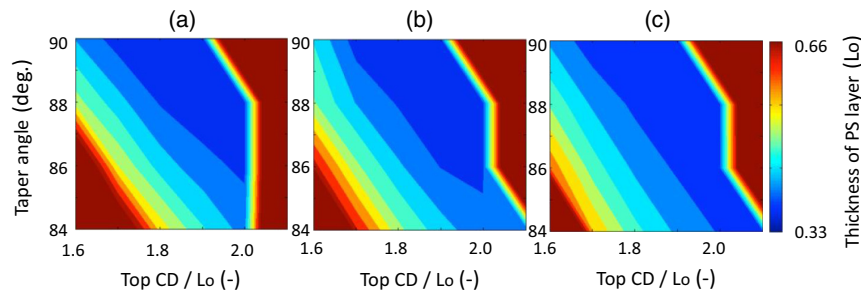
As  $\Lambda_s$  was increased from 10.0, the thickness of the PMMA layer was asymptotically decreased. We chose a sufficiently large value of  $\Lambda_s$  ( $= 100.0$ ) to stabilize the shape and thickness of the PMMA layer on the bottom surface.

### 3.3 Guide Hole Optimization

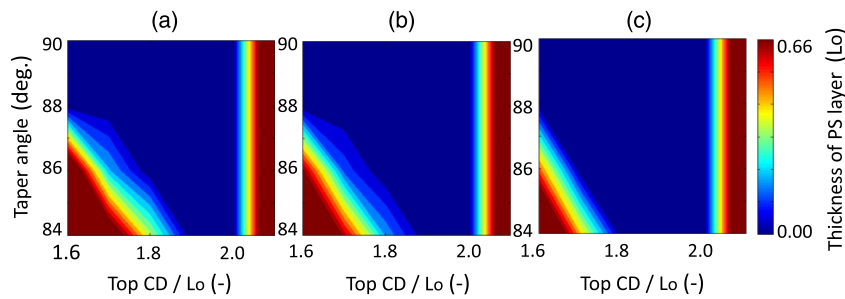
In order to decrease the thickness of the bottom PS layer, the shape of the guide hole was varied as follows: a taper angle from 84 to 90 (deg), a top CD from 1.6 to 2.1 [ $L_0$ ], and



**Fig. 7** Effect of the top CD on the morphology of PMMA-rich domain and the bottom PS layer. The taper angle and the hole height are fixed at 86 deg and 2.22 [ $L_0$ ], respectively. The top CD is increased from the left to the right: (a) 1.56, (b) 1.67, (c) 1.78, (d) 1.89, (e) 2.00, and (f) 2.11 [ $L_0$ ].



**Fig. 8** Contour plot of the thickness of the bottom PS layer with respect to the top CD (horizontal axis) and the taper angle (vertical axis). The three heights of the guide hole were examined: (a) 2.2, (b) 2.0, and (c) 1.8 [ $L_0$ ]. The color represents the thickness of the bottom PS layer.

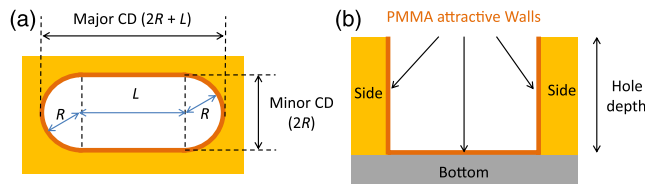


**Fig. 10** Contour plot of the thickness of the bottom PS layer with respect to the top CD (horizontal axis) and the taper angle (vertical axis), for the case of neutral bottom surface. The three heights of the guide hole were examined: (a) 2.2, (b) 2.0, and (c) 1.8  $[L_0]$ . The color represents the thickness of the bottom PS layer.

a depth hole from 1.8 to 2.2  $[L_0]$ . The surface parameters were set to be the ones obtained from the model calibrations (i.e.,  $\phi_s = 0.6$  and  $\Lambda_s = 100$ ).

Figure 7 shows an example of the top CD effect on the morphology of the cylindrical PMMA domain and the bottom PS layer. The thickness of the bottom PS layer was gradually decreased as that of the top CD was increased. At the top CD  $> 2.0 [L_0]$ , the cylindrical PMMA morphology was broken into the two smaller domains. Figure 8 illustrates the contour plots of the thickness of the bottom PS layer at three different heights of the guide hole. The area where the thickness of the bottom PS layer is relatively small (shown in blue color) becomes slightly larger with a decrease in height of the guide hole. It should be noted that, even at the hole height of 1.8  $[L_0]$ , the minimum thickness of the bottom PS layer is relatively large,  $\sim 0.3 [L_0]$ . It is difficult to reduce the thickness of the bottom PS layer by only varying the shape of the guide hole.

Next, the affinity of the bottom surface of the guide hole was modified from “PMMA attractive” to “neutral ( $\Lambda_s = 0$ ).” As shown in Fig. 9, with the neutral bottom surface, the cylindrical PMMA was able to touch the substrate, i.e., no PS layer in between. A similar effect of the surface affinity on the cylindrical morphology was reported in Ref. 8, where the simulations were performed by the SCFT. Figure 10 illustrates the process window for the neutral bottom surface. It is clearly seen that the “no PS residual layer” region (dark blue color) can exist in a relatively wide range, e.g., a taper angle from 88 to 90 (deg) at the top CD  $< \sim 2.0 [L_0]$ , or a top CD from  $\sim 1.8$  to  $\sim 2.0 [L_0]$  at a taper angle of 84 to 90 (deg).

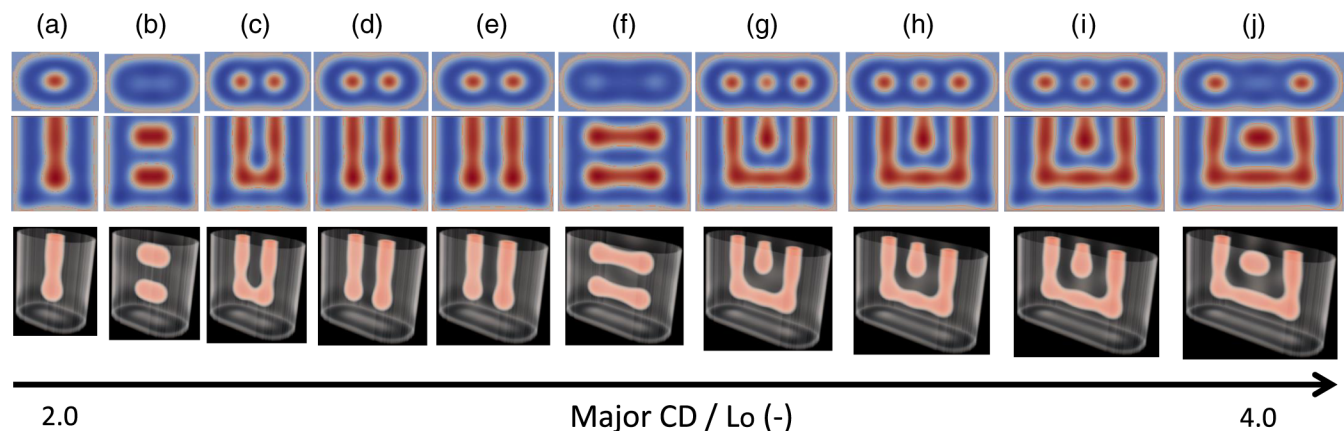


**Fig. 11** Parametric setup of the elliptical guide hole: (a) top-down image and (b) cross-sectional views.

### 3.4 Model Extensibility to Multiple Cylinders

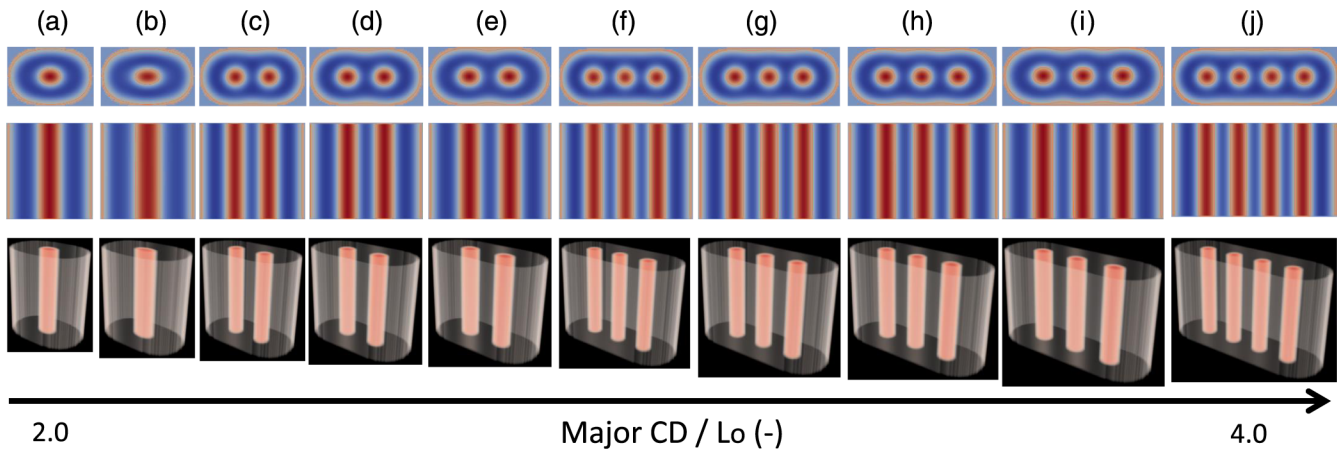
Using exactly the same model parameters, we have also investigated the phase-separated morphology of PS-b-PMMA in the elliptical holes. In experiments, multiple cylindrical PMMA domains were observed in the elliptical holes (e.g., Ref. 13). Figure 11 illustrates a schematic design of the elliptical guide hole. The sidewalls and bottom surface of the elliptical guide hole were assumed to be attractive to PMMA. The height and the minor CD of the guide hole were fixed at 2.2 and 1.3  $[L_0]$ , respectively. Only the major CD was changed from 2.0 to 4.0  $[L_0]$ , with an increment of 0.22  $[L_0]$ .

The simulation results are summarized in Fig. 12. The double cylinders were observed at a major CD of  $\sim 2.7$  to 2.9  $[L_0]$ . Interestingly, from the single to the double, and



**Fig. 12** Simulated phase-separated morphologies of PS-b-PMMA in the elliptical guiding hole: (top) top-down view, (middle) cross-sectional view, and (bottom) diagonal view of the simulated PMMA domain in the elliptical guide hole. The major CD is increased from (a) to (j), with the increment of 0.22  $[L_0]$ .





**Fig. 13** Effect of the neural bottom surface on the phase-separated morphologies of PS-b-PMMA in the elliptical guiding hole: (top) top-down view, (middle) cross-sectional view, and (bottom) diagonal view of the simulated PMMA domain in the elliptical guide hole. The major CD is increased from (a) to (j), with the increment of 0.22 [ $L_0$ ].

from the double to the triple, horizontal cylinders were formed. At the major CD of  $\sim 3.3$  to  $3.8 [L_0]$ , there were apparently triple cylindrical domains from the top view, but the two side ones were found to be connected via a horizontal cylindrical domain below the top surface. Such U-tube morphologies would be another type of defect, which could not be detected only by the top-down SEM images. In case the bottom surface was “neutral” instead of “PMMA attractive,” the morphological defects were eliminated and multiple cylindrical PMMA domains were in contact with the bottom surface (Fig. 13).

#### 4 Conclusions

In this study, we have applied the simplified model, called the Ohta–Kawasaki model, to find the optimal process conditions at which the thickness of the PS residual layer was minimized. Prior to the optimization, we demonstrated that the simplified model not only has a reasonable accuracy, but also an extremely short calculation time. Then, we calibrated the model parameters with the cross-sectional TEM images, and minimized the thickness of the PS residual layer by varying the guide hole shape (i.e., hole diameter, height, and taper angle). Our simulation results showed that it is difficult to eliminate the entire PS residual layer by only varying the guide hole shape. Alternatively, by changing the affinity of the bottom surface from “PMMA-attractive” to “neutral,” we could obtain a relatively wide, defect-free (i.e., no PS residual layer) process window. The same effect on the neutral bottom surface was observed for the case with an elliptical guide hole. Since our simulations are computationally inexpensive and scalable, they may also be feasible for large-scale simulations such as the hotspot (i.e., defects) analysis over a large area.

#### Acknowledgments

K. Y. and K. F. would like to acknowledge “Collaborative Research Program for Young Scientists of ACCMS and IIMC, Kyoto University,” for the use of the supercomputing facility. This research was partially supported by Japan

Society of the Promotion of Science (JSPS) KAKENHI with the Grant Number 25900237.

#### References

1. R. Gronheid et al., “Readying directed self-assembly for patterning in semi-conductor manufacturing,” *J. Photopolym. Sci. Technol.* **26**(6), 779–791 (2013).
2. Y. Seino et al., “Contact hole shrink process using graphoepitaxial directed self-assembly lithography,” *J. Micro/Nanolith. MEMS MOEMS.* **12**(3), 033011 (2013).
3. P. R. Delgadillo et al., “Defect source analysis of directed self-assembly process,” *J. Micro/Nanolith. MEMS MOEMS.* **12**(3), 033112 (2013).
4. C. Bencher et al., “Directed self-assembly defectivity assessment. Part II,” *Proc. SPIE* **8323**, 83230N (2012).
5. “The International Technology Roadmap for Semiconductors 2011 Ed. Lithography,” <http://www.itrs.net/Links/2011ITRS/2011Chapters/2011Lithography.pdf>.
6. K. Kodera et al., “Novel error mode analysis method for graphoepitaxial directed self-assembly lithography based on the dissipative particle dynamics method,” *Proc. SPIE* **8680**, 868015 (2013).
7. T. Nakano et al., “Dissipative particle dynamics study on directed self-assembly in holes,” *Proc. SPIE* **8680**, 86801J (2013).
8. N. Laachi et al., “The hole shrink problem: theoretical studies of directed self-assembly in cylindrical confinement,” *Proc. SPIE* **8680**, 868014 (2013).
9. K. Yamada, M. Nonomura, and T. Ohta, “Fddd structure in AB-type diblock copolymers,” *J. Phys.: Condens. Matter* **18**(32), L421–L427 (2006).
10. M. Pinna, X. Guao, and A. V. Zvelindovsky, “Diblock copolymers in a cylindrical pore,” *J. Chem. Phys.* **131**(21), 214902 (2009).
11. K. Yoshimoto and T. Taniguchi, “Large-scale simulations of directed self-assembly with simplified model,” *J. Photopolym. Sci. Technol.* **26**(6), 809–816 (2013).
12. T. Uneyama, “Density functional simulation of spontaneous formation of vesicle in block copolymer solutions,” *J. Chem. Phys.* **126**(11), 114902 (2007).
13. R. Tiron et al., “The potential of block copolymer’s directed self-assembly for contact hole shrink and contact multiplication,” *Proc. SPIE* **8680**, 868012 (2013).

**Kenji Yoshimoto** is a project associate professor at the Center for the Promotion of Interdisciplinary Education and Research, Kyoto University, Japan. He received a BS degree from Kyoto University in 1999, and a PhD degree (major: chemical engineering) from the University of Wisconsin—Madison in 2005. His current research interests include modeling and simulations, polymer physics, and regenerative medicine. He is a member of SPIE.

Biographies of the other authors are not available.